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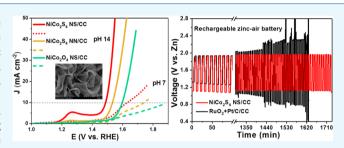
# Hierarchical Nickel—Cobalt Dichalcogenide Nanostructure as an Efficient Electrocatalyst for Oxygen Evolution Reaction and a Zn-Air **Battery**

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Supporting Information

ABSTRACT: A unique three-dimensional (3D) structure consisting of a hierarchical nickel-cobalt dichalcogenide spinel nanostructure is investigated for its electrocatalytic properties at benign neutral and alkaline pH and applied as an air cathode for practical zinc-air batteries. The results show a high oxygen evolution reaction catalytic activity of nickelcobalt sulfide nanosheet arrays grown on carbon cloth (NiCo<sub>2</sub>S<sub>4</sub> NS/CC) over the commercial benchmarking catalyst under both pH conditions. In particular, the NiCo<sub>2</sub>S<sub>4</sub> NS/CC air cathode shows high discharge capacity,



a narrow potential gap between discharge and charge, and superior cycle durability with reversibility, which exceeds that of commercial precious metal-based electrodes. The excellent performance of NiCo<sub>2</sub>S<sub>4</sub> NS/CC in water electrolyzers and zinc-air batteries is mainly due to highly exposed electroactive sites with a rough surface, morphology-based advantages of nanosheet arrays, good adhesion between NiCo<sub>2</sub>S<sub>4</sub> and the conducting carbon cloth, and the active layer formed of nickel-cobalt (oxy)hydroxides during water splitting. These results suggest that NiCo<sub>2</sub>S<sub>4</sub> NS/CC could be a promising candidate as an efficient electrode for high-performance water electrolyzers and rechargeable zinc-air batteries.

#### INTRODUCTION

Splitting water into pure hydrogen and oxygen to generate sustainable green hydrogen energy has been intensively studied in recent years, which can replace fossil fuel use. 1,2 However, the efficiency of water splitting has so far been limited by the lack of sustainable catalysts toward the oxygen evolution reaction (OER) that can accelerate the kinetics.<sup>3–5</sup> So far, IrO<sub>x</sub> and RuO2 are the best-known OER catalysts, although their high cost and scarcity limit their widespread use. 4 Meanwhile, some promising attempts have been devoted to developing an efficient nonprecious metal OER catalyst under alkaline conditions. However, an almost harsh alkaline medium presents severe corrosion and related environmental issues.<sup>6,7</sup> In this regard, someday, the splitting of water at neutral pH from ocean or river would be the target goal to satisfy renewable future hydrogen energy.4 It is thus highly required to develop efficient OER electrocatalysts that can operate in both alkaline and neutral media for overall water splitting even though it is relatively tough searching for those catalysts.

Nowadays, various nonprecious transition metal-based catalysts are being explored, for example, transition metals, transition-metal oxides, 2,10 chalcogenides, 11-14 phosphides, 15-17 hydroxides/oxyhydroxides, 18,19 carbides, 20 boridae 21 and 1 ides,<sup>21</sup> and so on. Although lots of established catalysts have been reported concerning their excellent OER activity under alkaline conditions, only a few of them could still maintain their catalytic activity in neutral media. Cai et al. reported that

the amorphous cobalt sulfide porous nanocubes showed a low OER onset potential of 1.5 V, comparable to that of RuO<sub>2</sub> (1.49 V).<sup>22</sup> However, a still substantial overpotential of 570 mV is needed to generate 4.59 mA cm<sup>-2</sup> in phosphate-buffered solutions (PBSs; pH 7.0), whereas it could generate 10 mA cm<sup>-2</sup> current density at 290 mV in 1 M KOH (pH 14.0). Similarly, sulfur-incorporated NiFe<sub>2</sub>O<sub>4</sub> nanosheets (NSs) on nickel foam (S-NiFe<sub>2</sub>O<sub>4</sub>/NF) developed by Liu et al. exhibited a remarkably enhanced water-splitting performance for both OER and hydrogen evolution reaction (HER) as a bifunctional electrode under both alkaline and neutral conditions.<sup>23</sup> The S-NiFe<sub>2</sub>O<sub>4</sub>/NF still requires 1.921 V to deliver 10 mA cm<sup>-2</sup> in 1 M PBS (pH 7.4) for overall water splitting in three electrode systems, mostly occurring during OER with an overpotential of 494 mV. As air cathode catalysts for Zn-air batteries, Prabu et al., demonstrated a highly active one-dimensional structure of a spinel NiCo2O4 catalyst in rechargeable Zn-air batteries and Li-O<sub>2</sub> batteries. <sup>24,26</sup> Recently, Meng et al. constructed Co<sub>0.85</sub>Se nanocrystals in situ coupled with N-doped carbon with a metal-nitrogencarbon (M-N-C) structure and short diffusion pathways for the transport of electron/ion to improve the Zn-air battery performance. 25,27 For instance, Wu et al. reported zinc cobalt

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sulfide, the nanoneedle (NN) arrays grown on the carbon fiber paper electrode catalyst, which enables the Zn–air battery operation with an overpotential of 0.85 V and a long cycle life time of up to 200 cycles at 10 mA cm $^{-2}$  as well as comparable water-splitting performance. Wang et al. proposed  ${\rm Co_3FeS_{1.5}(OH)_6}$  hydroxysulfides serving as a superb air electrode catalyst with a low overpotential of 0.84 V and prolonged cyclability over 36 h test for 108 cycles at 2 mA cm $^{-2.29}$ 

In pursuit of high electrochemical performance in water electrolyzers and for a Zn-air battery application, the spinel bimetallic sulfide NiCo<sub>2</sub>S<sub>4</sub> with abundant redox chemistry has been considered to be the most promising electrochemically active material, which exhibits 2 orders of magnitude larger than that of NiCo<sub>2</sub>O<sub>4</sub> and ~10<sup>4</sup> times better electric conductivity than conventional single-metal compounds. 12,30 Moreover, the stable spinel structures of bimetallic sulfide with a formula of AB<sub>2</sub>S<sub>4</sub> possess plentiful exposed edge sites, leading to a higher electrochemical activity. Therefore, it has been widely applied for supercapacitors, Li-ion batteries, and a counter electrode for dye-sensitized solar cells as well as in water electrolyzers.<sup>31</sup> For example, in our group, NiCo<sub>2</sub>S<sub>4</sub> nanowire arrays were directly grown on 3D Ni foam (NiCo<sub>2</sub>S<sub>4</sub> NW/NF) as a water-splitting catalyst and applied in an alkaline water electrolyzer. Because of its intrinsic properties, large surface area, and well-separated NW structures, NiCo<sub>2</sub>S<sub>4</sub> NW/ NF afforded continuous water-splitting reaction of generating hydrogen and oxygen gas at a cell voltage of only 1.63 V to generate 10 mA cm<sup>-2</sup> current density.<sup>32</sup> Ma et al. developed 3D networked porous NiCo<sub>2</sub>S<sub>4</sub> nanoflakes on NF, which can offer more exposed active sites and easy transport of electrons and ions, thereby leading to significantly improved HER activity and stability.<sup>33</sup> The outstanding OER performance of the spinel bimetallic sulfide NiCo<sub>2</sub>S<sub>4</sub> has also become a promising application for rechargeable Zn-air batteries involving reversible OER and ORR.

Also, hybridizing NiCo<sub>2</sub>S<sub>4</sub> with the 3D structure of NS or NN with a carbon cloth (CC) (NiCo<sub>2</sub>S<sub>4</sub> NS/CC or NN/CC) substrate by an in situ growth hydrothermal approach can be an efficient way to enhance the robustness of the electrode. It could also help to minimize the agglomeration of NiCo<sub>2</sub>S<sub>4</sub> nanostructures and the detachment during long-term operation, and make faster ion/electron kinetics. Also, the advantages of CC, such as flexibility, high conductivity, and corrosion/dissolution resistivity might lead to enhanced catalytic activity and stability in a wide pH range.<sup>34</sup> Meanwhile, tuning the nanostructure and the morphology, as well as porosity, can be another promising strategy to produce numerous exposed catalytic active sites on the catalyst surface.<sup>17</sup>

On the basis of our knowledge, we describe the physical and electrochemical properties of NiCo<sub>2</sub>S<sub>4</sub> NS/CC as a highly active OER catalyst in both neutral and alkaline media, which have not been thoroughly investigated so far. Mainly, this is the first time a bimetallic sulfide, the NiCo<sub>2</sub>S<sub>4</sub>-based material, is reported to catalyze OER under a neutral condition. The catalytic performance of NiCo<sub>2</sub>S<sub>4</sub> NS/CC is remarkably enhanced compared to the recent reports, particularly under a neutral condition. The NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrocatalyst exhibits the lowest OER overpotentials of 260 and 402 mV to generate 10 mA cm<sup>-2</sup> in alkaline and neutral media, respectively. Specifically, it exhibits a low Tafel slope of 123 mV dec<sup>-1</sup> and a high turnover frequency (TOF) of 8.17  $\times$ 

 $10^{-3}~\rm s^{-1}$  at 1.63 V applied potential to drive 10 mA cm $^{-2}$  current density under neutral conditions, confirming superior intrinsic activity with a substantial electrochemical active surface area (ECSA) of NiCo<sub>2</sub>S<sub>4</sub> NS/CC compared with commercial RuO<sub>2</sub>/CC and other previously reported OER electrocatalysts. In addition, the constructed NiCo<sub>2</sub>S<sub>4</sub> NS/CC air cathode for primary and rechargeable Zn–air batteries exhibits high discharge capacity, a narrow overall overpotential, and a long cycling life time exceeding the benchmark for precious metal-based electrodes.

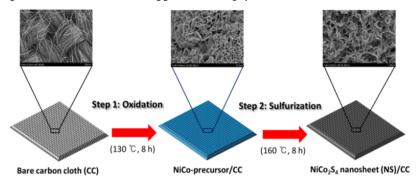
# **EXPERIMENTAL METHODS**

Material Synthesis. Cobalt nitrate hexahydrate [Alfa Aesar, Co(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O], nickel nitrate hexahydrate [Sigma-Aldrich, Ni(NO<sub>3</sub>)<sub>2</sub>·6H<sub>2</sub>O], urea (Sigma-Aldrich, CH<sub>4</sub>N<sub>2</sub>O), and sodium sulfide hydrate (Sigma-Aldrich, Na<sub>2</sub>S·xH<sub>2</sub>O) were used to synthesize the electrodes. A piece of CC (NARA CELL-TECH, 0.7 cm × 0.7 cm) was utilized with further treatment with ethanol. The NiCo<sub>2</sub>S<sub>4</sub> NSs grown on CC (NiCo<sub>2</sub>S<sub>4</sub> NS/CC) were prepared through a two-step hydrothermal process. A nickel nitrate of 0.004 M and cobalt nitrate of 0.008 M was dissolved in 120 mL of deionized water; further, 0.012 M urea was added. The obtained solution was transferred into a Teflon-lined stainless steel autoclave of 200 mL capacity, and a piece of CC was immersed in the solution. The autoclave was heated at 120 °C for 8 h in an electric oven. After the first step, the electrode was washed with deionized water several times to eliminate unreacted residues. Consequently, sodium sulfide flakes were dissolved in deionized water to prepare a 0.2 M sulfide solution for a sulfurization process. This sulfur-containing solution was again transferred to the autoclave and heated at 160 °C for 8 h in an electric oven. After cooling down to room temperature naturally, we washed the synthesized electrode several times with ethanol and deionized water, followed by the drying step in the vacuum oven at 60 °C overnight. For comparison, NiCo<sub>2</sub>S<sub>4</sub> NN arrays on CC (NiCo<sub>2</sub>S<sub>4</sub> NN/CC) were prepared by changing the temperature of the first hydrothermal step from 120 to 130 °C and maintaining the heating time of 8 h. To synthesize NiCo2O4 with NSs morphology, which is grown on CC (NiCo<sub>2</sub>O<sub>4</sub> NS/CC), the electrode after the first step of the hydrothermal growth process was annealed at 450 °C for 2 h in an air atmosphere.

**Microstructural Characterizations.** The morphology and element compositions were studied on a field-emission scanning electron microscopy (FE-SEM, Hitachi-S4800, 3 kV) system equipped with a Horiba Scientific energy dispersive spectrometer and using transmission electron microscopy (TEM, Hitachi HF-3300, 300 kV). The crystal structures of all catalysts were examined by powder X-ray diffraction (XRD, Rigaku MiniFlex600). The composition of the catalyst was studied using X-ray photoelectron spectroscopy (XPS, Thermo-Scientific ESCALAB 250Xi).

**Electrochemical Measurements.** For electrochemical measurements, the OER catalytic performance was evaluated by linear sweep voltammetry (LSV) with a low scan rate of 1 mV s<sup>-1</sup> in an electrolyte of 1 M KOH and a PBS without purging oxygen. The OER performance was evaluated in a three-electrode configuration directly using synthesized electrodes such as NiCo<sub>2</sub>S<sub>4</sub> NS/CC, NiCo<sub>2</sub>S<sub>4</sub> NN/CC, and NiCo<sub>2</sub>O<sub>4</sub> NS/CC as a working electrode (1.0 mg cm<sup>-2</sup>), a saturated calomel electrode (SCE) as a reference electrode, and a Pt wire as the counter electrode. Similarly, commercial

Scheme 1. Schematic Preparation Process of Self-Supported NiCo<sub>2</sub>S<sub>4</sub> NS/CC Nanostructures



RuO $_2$  cast onto CC was used as a working electrode (1.4 mg cm $^{-2}$ ), Pt wire as a counter electrode, and SCE as a reference electrode. The potentials reported were converted to the reversible hydrogen electrode (RHE). All electrochemical results were iR-corrected, considering the ohmic resistance from the electrolyte. The current densities presented in this paper are normalized concerning the geometric surface area of the electrode. The cyclic voltammetry (CV) was performed in N $_2$ -saturated 1 M KOH at room temperature with a scan rate of 10 mV s $^{-2}$ . Electrochemical impedance spectroscopy (EIS) was performed within a frequency range of 0.01 Hz to 0.1 MHz.

The ECSA is calculated by following an established methodology reported in the literature. In detail, through the cyclic voltammogram obtained in a non-faradaic region at various scan rates (1, 2.5, 5, 10, 20, and 50 mV s<sup>-1</sup>), double-layer capacitance ( $C_{\rm dl}$ ) can be estimated. By plotting the anodic and cathodic current densities against the scan rate, the obtained linear slope value is  $C_{\rm dl}$ . Finally, the ECSA can be obtained from the following equation:

$$ECSA = C_{dl}/C_{s}$$

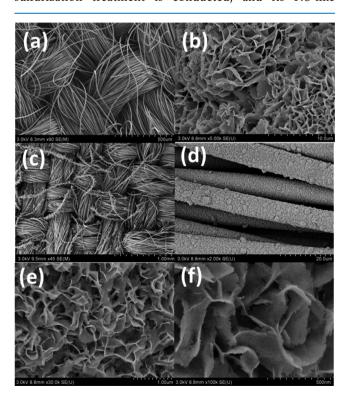
 $C_{\rm s}$  denotes the specific capacitance of a flat, smooth surface of the electrode material, which is assumed to be 26  $\mu{\rm F~cm}^{-2}$  for Ni- and Co-containing materials.

Zinc-Air Battery Fabrication and Testing. For full-cell zinc-air battery evaluation, the as-synthesized NiCo<sub>2</sub>S<sub>4</sub> NS/ CC or commercial catalyst of RuO<sub>2</sub> + Pt/C/CC was used as an air-breathing cathode (1.54 cm<sup>2</sup>). Typically, the NiCo<sub>2</sub>S<sub>4</sub> NS/ CC air cathode was prepared with a loading amount of 1.0 mg cm<sup>-2</sup>. The commercial catalyst-based cathode was fabricated by coating with 2 mg of RuO2 and 40 wt % Pt/C on CC to achieve a loading of 1.30 mg cm<sup>-2</sup>. A polished zinc plate with a thickness of 0.1 mm, 6 M KOH solution with 0.2 M zinc acetate, and a Whatman glass microfiber filter membrane were prepared as an anode, an electrolyte, and a separator, respectively, to assemble the zinc-air battery with a coin cell (MTI Korea) configuration. The specific capacity and the charge-discharge curves were reported with a battery analyzer (BST8-3), which consumed ambient air. The discharge capacity of the primary zinc-air cell was normalized to the consumed mass of Zn metal, whereas the current density was normalized to the area of the electrode.

## ■ RESULTS AND DISCUSSION

The hierarchical bimetallic sulfide NS arrays/CC hybrids were developed using an in situ two-step hydrothermal method, as graphically represented in Scheme 1. In the first step of the hydrothermal process, nickel nitrate hexahydrate and cobalt

nitrate hexahydrate in a stoichiometric ratio were dissolved in deionized water and then urea was added. The solution was transferred into an autoclave and heated at 120 °C for 8 h, forming cobalt-nickel carbonate hydroxide hydrate NS arrays on a CC substrate. Subsequently, after oxidation reaction, a solution of sodium sulfide flakes dissolved in deionized water was prepared for the next sulfurization process. The anion exchange reaction from Co<sub>3</sub><sup>2-</sup>/OH<sup>-</sup> anions to S<sup>2-</sup> anions occurred at 160 °C for 6 h, thereby leading the complete phase transformation from cobalt-nickel carbonate hydroxide hydrate to nickel-cobalt sulfide on the CC. The morphology and composition of NiCo<sub>2</sub>S<sub>4</sub> NS/CC were studied by FE-SEM and TEM. The bare CC consists of interconnected fibers with a smooth surface, as shown in Figure 1a. After the first hydrothermal reaction, numerous NS arrays are stacked onto the surface of CC with a rough surface shown in Figures 1b and S1. Consequently, the second hydrothermal process for sulfurization treatment is conducted, and its NS-like



**Figure 1.** FE-SEM images of (a) bare CC, (b) NiCo-precursor/CC after the first step in the hydrothermal process, (c–f) NiCo<sub>2</sub>S<sub>4</sub> NS arrays grown on the surface of CC at different magnifications.

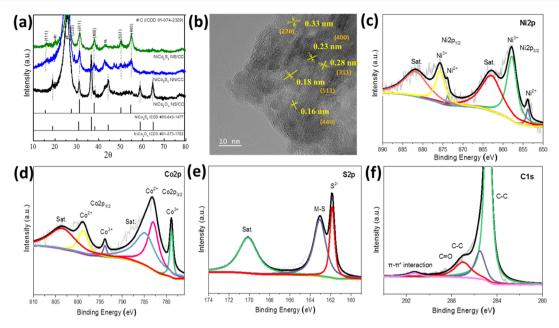


Figure 2. (a) XRD patterns of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, NiCo<sub>2</sub>S<sub>4</sub> NN/CC, and NiCo<sub>2</sub>O<sub>4</sub> NS/CC catalysts. (b) High-resolution TEM images of NiCo<sub>2</sub>S<sub>4</sub> NS/CC. High-resolution XPS deconvolution spectra of the NiCo<sub>2</sub>S<sub>4</sub> NS/CC catalyst for (c) Ni 2p, (d) Co 2p, (e) S 2p, and (f) C 1s.

morphology perpendicular to a substrate with a rough surface still preserves its architecture (Figure 1e). From the top view of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, as shown in Figure 1c,d, the NiCo<sub>2</sub>S<sub>4</sub> NSs that are interconnected with each other and form a porous architecture with submicron-size pores, providing ample space for the fast diffusion of redox ions during the reaction, uniformly cover the surface of CC. For comparison, the needle-like shape of NiCo<sub>2</sub>S<sub>4</sub> vertically grown on the surface of CC with the average length of 3  $\mu$ m is also synthesized to understand the effect of morphology on catalytic activities in both alkaline and neutral medium (Figure S2a-c). Also, the NiCo<sub>2</sub>O<sub>4</sub> NS/CC was fabricated to have the same morphological characteristics as NiCo<sub>2</sub>S<sub>4</sub> NS/CC (Figure S2d-f).

The XRD patterns of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, NiCo<sub>2</sub>S<sub>4</sub> NN/CC, and NiCo2O4 NS/CC are presented in Figure 2a. After oxidation reaction in the first step of the hydrothermal process, nickel-cobalt carbonate hydroxide hydrate NS arrays are uniformly formed on the CC, as shown in Figure S1 (ICDD 00-040-0216). During the sulfurization in the second hydrothermal step, the nickel-cobalt carbonate hydroxide hydrate phase is completely transformed into spinel nickel-cobalt sulfide without destroying the original nanostructures. The diffraction peaks of NiCo<sub>2</sub>S<sub>4</sub> NS/CC and NiCo<sub>2</sub>S<sub>4</sub> NN/CC at  $16.2^{\circ}$ ,  $26.7^{\circ}$ ,  $31.4^{\circ}$ ,  $38.1^{\circ}$ ,  $50.3^{\circ}$ , and  $55.1^{\circ}$  are assigned to the (111), (220), (311), (400), (511), and (440) planes of cubicphase NiCo<sub>2</sub>S<sub>4</sub>, respectively (ICDD 00-043-1477). The highresolution TEM image in Figure 2b reveals the interplanar distance of 0.28, 0.23, 0.16, 0.33, and 0.18 nm, corresponding to the (311), (400), (440), (220), and (511) planes of NiCo<sub>2</sub>S<sub>4</sub>, respectively, confirming the successful formation of NiCo<sub>2</sub>S<sub>4</sub>. The formation of NiCo<sub>2</sub>S<sub>4</sub> NS/CC is confirmed based on the energy-dispersive X-ray (EDX) using TEM, which shows that the atomic percentages of nickel, cobalt, and sulfur are 15.2, 30.7, and 54.1 at. %, respectively, as shown in Figure S3. Note that the two characteristic peaks at 26° and 43° for all prepared electrodes are attributed to the CC (ICDD 01-074-2329). The diffraction peaks of the NiCo<sub>2</sub>O<sub>4</sub> NS/CC catalyst obtained after the first step in the hydrothermal

process and heat treatment are consistent with the standard pattern of  $NiCo_2O_4$  (ICDD 01-073-1702). Meanwhile, the NN-like morphology of  $NiCo_2S_4$  NN/CC can be confirmed from the TEM elemental mapping images shown in Figure S4.

To further characterize the chemical composition of electrodes, the XPS analysis is carried out, and results are given in Figure 2c-f. As shown in Figure 2c, the Ni 2p spectrum consists of two spin-orbit doublets of Ni<sup>2+</sup> and Ni<sup>3+</sup> including Ni<sup>2+</sup> at 853.8 eV for Ni  $2p_{3/2}$  and 873.8 eV for Ni  $2p_{1/2}$ , and Ni<sup>3+</sup> at 857.7 eV for Ni  $2p_{3/2}$  and 875.6 eV for Ni  $2p_{1/2}$ . The XPS spectrum of Co 2p (Figure 2d) contains well-resolved peaks of Co<sup>2+</sup> 2p<sub>3/2</sub>, Co<sup>3+</sup> 2p<sub>3/2</sub>, Co<sup>2+</sup>  $2p_{1/2}$ , and  $Co^{3+}$   $2p_{1/2}$  at 798.8, 793.9, 783.0, and 778.9 eV, respectively, implying the co-presence of Co<sup>2+</sup> and Co<sup>3+</sup> species in NiCo<sub>2</sub>S<sub>4</sub> NS/CC.<sup>23,35</sup> The S 2p XPS spectrum (Figure 2e) shows two peaks at 163.0 and 161.8 eV, which are assigned to metal-sulfur bonds and the low coordination state sulfur ion that exists at the surface of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, respectively, with the satellite peak appearing at 170.1 eV in Figure 2e. 35,36 The C 1s spectrum of NiCo<sub>2</sub>S<sub>4</sub> NS/CC is deconvoluted into four peaks located at 284.8, 285.5, 287.0, and 291.3 eV, which correspond to the C 1s orbital of C-C (sp<sup>2</sup>), C-C (sp), C-O, and  $\pi-\pi$  interactions, respectively (Figure 2f). The additional  $\pi - \pi$  interaction indicates the strong interactions between NiCo<sub>2</sub>S<sub>4</sub> NSs arrays and the CC, which can minimize contact resistance to generate a direct electron pathway.<sup>30</sup> The binding energy values of Ni 2p, Co 2p, and S 2p are matched well with the previous reports on NiCo<sub>2</sub>S<sub>4</sub>-based materials.

The OER catalytic activity of NiCo<sub>2</sub>S<sub>4</sub> NS/CC was first evaluated with a three-electrode setup using a low scan rate of 1 mV s<sup>-1</sup> to eliminate the capacitive current effects in alkaline solution (1 M KOH, pH = 14). For comparison, NiCo<sub>2</sub>S<sub>4</sub> NN/CC, NiCo<sub>2</sub>O<sub>4</sub> NS/CC, bare CC, and RuO<sub>2</sub>/CC benchmarking OER catalysts were also tested under the same condition. All the synthesized electrodes in this work are directly used as free-standing oxygen-evolving electrodes, including conventional RuO<sub>2</sub>/CC, to avoid possible influenc-

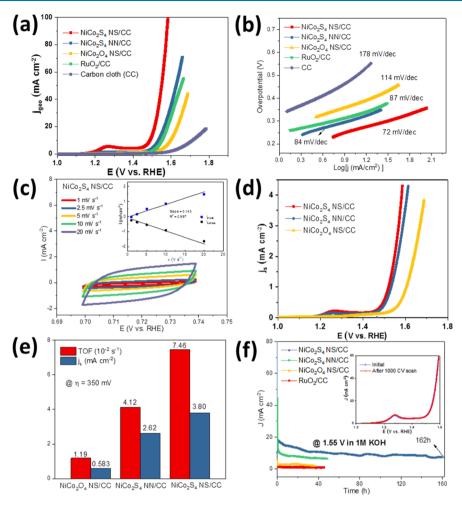


Figure 3. OER in alkaline media (1 M KOH, pH = 14): (a) LSV polarization curves for OER and (b) Tafel plots of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, along with NiCo<sub>2</sub>S<sub>4</sub> NN/CC, NiCo<sub>2</sub>O<sub>4</sub> NS/CC, RuO<sub>2</sub>/CC, and CC electrodes for comparison. (c) Cyclic voltammogram measured in a non-faradaic region at various scan rates for the NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrode. The inset shows the plot of anodic and cathodic charging current density vs different scan rates. (d) Current density based on intrinsic catalytic activity vs voltage curves. (e) TOF and the specific (intrinsic) activities of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, NiCo<sub>2</sub>S<sub>4</sub> NN/CC, and NiCo<sub>2</sub>O<sub>4</sub> NS/CC electrodes at η = 350 mV. (f) Time dependence of the current density for NiCo<sub>2</sub>S<sub>4</sub> NS/CC at a fixed potential of 1.55 V for 162 h.

ing factors. The LSV polarization curves and Tafel plots for all samples are revealed, as shown in Figure 3a,b. The NiCo<sub>2</sub>S<sub>4</sub> NS/CC exhibits a superior catalytic activity toward OER with a low onset potential of only 180 mV. Moreover, the overpotential of 260 mV is required to generate 10 mA cm<sup>-2</sup>, which is smaller than that of NiCo<sub>2</sub>S<sub>4</sub> NN/CC (316 mV), NiCo<sub>2</sub>O<sub>4</sub> NS/CC (368 mV), RuO<sub>2</sub>/CC (322 mV), and bare CC (484 mV). Also, it requires only a 280 mV overpotential to afford 10 mA cm<sup>-2</sup> when the scan rate is 5 mV s<sup>-1</sup> (Figure S5). It is lower than those of many other reported nonprecious metal-based OER electrocatalysts tested under 1 M KOH conditions, such as NiCo<sub>2</sub>S<sub>4</sub> NWs/ graphdiyne foam (300 mV),  $NiCo_2S_4/NF$  (306 mV)<sup>-</sup>,  $NiCo_2S_4$  NAs/CC (310 mV), and so on.  $^{22-24,30,32,37-40}$  A detailed comparison is further summarized in Table S1. The cyclic voltammograms for NiCo<sub>2</sub>S<sub>4</sub> NS/CC and NiCo<sub>2</sub>O<sub>4</sub> NS/ CC in the potential region from 1.0 to 1.8 V show a broad peak at 1.36 and 1.22 V, indicating the redox behavior of Ni<sup>2+</sup> and Ni<sup>3+</sup>, respectively (Figure S7). The catalytic kinetics of OER is evaluated by Tafel plots in alkaline medium. The Tafel slope of NiCo<sub>2</sub>S<sub>4</sub> NS/CC is 72 mV dec<sup>-1</sup>, which is lower than that of all other electrodes, such as RuO<sub>2</sub>/CC (87 mV dec<sup>-1</sup>), NiCo<sub>2</sub>S<sub>4</sub> NN/CC (84 mV dec<sup>-1</sup>), NiCo<sub>2</sub>O<sub>4</sub> NS/CC (114 mV

dec<sup>-1</sup>), and bare CC (178 mV dec<sup>-1</sup>), indicating a more favorable rate of OER at the NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrode. The favorable kinetics of OER on NiCo<sub>2</sub>S<sub>4</sub> NS/CC is also supported by EIS analysis to measure the charge transfer resistance during OER (Figure S6). The charge transfer resistance of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, which forms NSs morphology is 8.94  $\Omega$  at 1.5 V versus RHE, smaller than that of NiCo<sub>2</sub>S<sub>4</sub> NN/CC with NN-like architectures (16.3  $\Omega$ ). In contrast, NiCo<sub>2</sub>O<sub>4</sub> NS/CC shows at least five times higher charge transfer resistance than that of NiCo<sub>2</sub>S<sub>4</sub> NS/CC and NiCo<sub>2</sub>S<sub>4</sub> NN/CC under the same applied potential because of the lower electrical conductivity of NiCo<sub>2</sub>O<sub>4</sub> NS/CC.

To better understand the different OER catalytic activities of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, including NiCo<sub>2</sub>S<sub>4</sub> NN/CC and NiCo<sub>2</sub>O<sub>4</sub> NS/CC catalysts, the ECSA and roughness factor (RF) of all electrodes are determined to estimate the real catalytic activities in the same pH condition. It can be easily calculated based on the double-layer capacitance ( $C_{\rm dl}$ ) through CV in a non-faradaic region at different scan rates of 1, 2.5, 5, 10, 20, and 50 mV s<sup>-1</sup> (Figure 3c). The NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrode shows over twofold higher ECSA value of 5.5 mF cm<sup>-2</sup> than that of NiCo<sub>2</sub>S<sub>4</sub> NN/CC (2.0 mF cm<sup>-2</sup>) and NiCo<sub>2</sub>O<sub>4</sub> NS/CC (2.3 mF cm<sup>-2</sup>), respectively (Figure S8). This indicates that

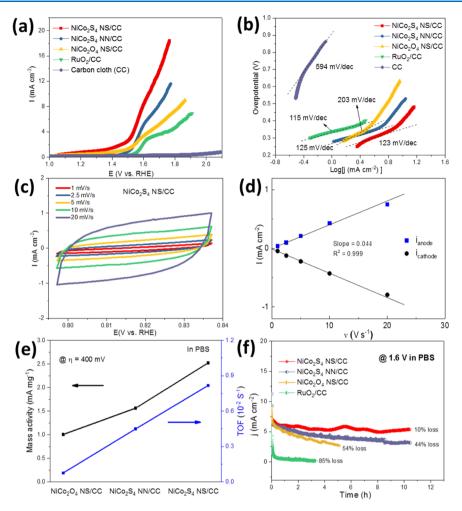


Figure 4. OER in neutral media (phosphate buffer, pH = 7): (a) LSV polarization curves for OER. (b) Corresponding Tafel plots of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, along with NiCo<sub>2</sub>S<sub>4</sub> NN/CC, NiCo<sub>2</sub>O<sub>4</sub> NS/CC, RuO<sub>2</sub>/CC, and CC electrodes for comparison. (c) Cyclic voltammogram measured in a non-faradaic region at various scan rates for the NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrode. (d) Anodic and cathodic current density vs scan rate plot of NiCo<sub>2</sub>S<sub>4</sub> NS/CC. (e) Mass activities and TOF of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, NiCo<sub>2</sub>S<sub>4</sub> NN/CC, and NiCo<sub>2</sub>O<sub>4</sub> NS/CC electrodes in PBS. (f) Time dependence of the current density for NiCo<sub>2</sub>S<sub>4</sub> NS/CC at a fixed potential of 1.6 V for 11 h.

plenty of catalytically active sites for OER might form on NiCo<sub>2</sub>S<sub>4</sub> NS/CC.

The surface roughness for all electrodes was also calculated by dividing the estimated ECSA to the geometric area of the electrode, and the values of 27.5, 10, and 11.5 were achieved for each electrode, NiCo<sub>2</sub>S<sub>4</sub> NS/CC, NiCo<sub>2</sub>S<sub>4</sub> NN/CC, and NiCo<sub>2</sub>O<sub>4</sub> NS/CC, respectively. On the basis of these results, 2D NS architecture arrays can offer larger space and have a rougher surface; hence, they lead to more electrochemical active sites on the catalyst surface. As a result, the excellent electrocatalytic performances of the NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrode can be partially ascribed to the high ECSA and consequently highly exposed active sites.

We further calculate the TOF, which could provide the intrinsic OER catalytic activities of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, NiCo<sub>2</sub>S<sub>4</sub> NN/CC, and NiCo<sub>2</sub>O<sub>4</sub> NS/CC electrodes. The TOF of various electrocatalysts was derived using the following equation:

$$TOF = \frac{J_{geo} \times A}{4 \times F \times m}$$

J is the geometric current density at a specific overpotential. A denotes the geometric area of the electrode. The number of

electrons consumed for generating 1 mol of  $O_2$  from water is 4. F is the Faraday constant value of 96 485 C mol<sup>-1</sup>. m denotes the mole numbers of active materials. On the basis of this calculation, at an overpotential of 350 mV, the TOF of NiCo<sub>2</sub>S<sub>4</sub> NS/CC and NiCo<sub>2</sub>S<sub>4</sub> NN/CC is calculated as 7.46 ×  $10^{-2}$  and  $4.12 \times 10^{-2}$  mol  $O_2$  s<sup>-1</sup>, respectively (Figure 3e). In sharp contrast, NiCo<sub>2</sub>O<sub>4</sub> NS/CC has the lowest TOF value of  $1.19 \times 10^{-2}$  s<sup>-1</sup>. Moreover, it is almost fourfold higher than that of the IrO<sub>x</sub> catalyst (0.89 ×  $10^{-2}$  s<sup>-1</sup>), indicating that the NiCo<sub>2</sub>S<sub>4</sub> NS/CC is highly efficient toward OER. <sup>42</sup>

The specific activity of catalysts with different surface areas or loading is calculated with current normalization by the catalyst RF. NiCo<sub>2</sub>S<sub>4</sub> NS/CC can deliver a high specific current density of 2.67 mA cm<sup>-2</sup>, whereas NiCo<sub>2</sub>O<sub>4</sub> NS/CC can produce only 0.96 mA cm<sup>-2</sup> at the same overpotential of 350 mV. Even though their morphology appears to be similar, that is, the density and the distribution of NiCo<sub>2</sub>S<sub>4</sub> NSs on the CC are virtually the same as those of NiCo<sub>2</sub>O<sub>4</sub> NSs on CC, NiCo<sub>2</sub>S<sub>4</sub> NS/CC outperforms NiCo<sub>2</sub>O<sub>4</sub> NS/CC. This result shows that NiCo<sub>2</sub>S<sub>4</sub> NS arrays on CC have a higher intrinsic OER catalytic activity than that of NiCo<sub>2</sub>O<sub>4</sub> NS arrays on CC (Figure 3d), which can be explained by the difference in the crystal structure of NiCo<sub>2</sub>S<sub>4</sub> and NiCo<sub>2</sub>O<sub>4</sub>. NiCo<sub>2</sub>S<sub>4</sub> that

formed closely packed arrays of large  $S^{2-}$  anions with nickel and cobalt metal cations in different oxidation states occupying the tetrahedral and octahedral sites, respectively, possesses more octahedral active sites of Co(III) concerning NiCo<sub>2</sub>O<sub>4</sub>, which has smaller anions of O<sup>2-</sup> in the spinel structure. <sup>32,43</sup> On the basis of the previous literature,  $\sigma^*$  orbital (e<sub>g</sub>) occupation—related metal cations at octahedral sites are mostly coordinated with electrocatalytic activities. <sup>44</sup> In view of this point, NiCo<sub>2</sub>S<sub>4</sub> NS/CC might afford better OER intrinsic activity compared to NiCo<sub>2</sub>O<sub>4</sub> NS/CC.

The electrocatalytic activity toward the OER of NiCo<sub>2</sub>S<sub>4</sub> NS/CC is also evaluated in PBS (pH = 7) as well as control samples as shown in Figure 4a. Similar to the OER activity trend in alkaline media, the NiCo<sub>2</sub>S<sub>4</sub> NS/CC catalyst exhibits the highest OER performance compared with other electrodes. Surprisingly, NiCo<sub>2</sub>S<sub>4</sub> NS/CC requires only 321 and 402 mV to afford 5 and 10 mA cm<sup>2</sup>, respectively. However, RuO<sub>2</sub>/CC as a state-of-the-art OER catalyst needs an extremely large overpotential of 700 mV to deliver 5 mA cm<sup>-2</sup> current density. At the same time, NiCo<sub>2</sub>S<sub>4</sub> NN/CC and NiCo<sub>2</sub>O<sub>4</sub> NS/CC require at least 368 and 460 mV to produce 5 mA cm<sup>-2</sup>, respectively. Bare CC shows negligible OER performance. The catalytic activity of NiCo<sub>2</sub>S<sub>4</sub> NS/CC for OER in neutral media is exceptional compared to that of many electrodes reported recently, such as  $CoS_{4.6}O_{0.6}$  ( $\eta = 570 \text{ mV for 5 mA cm}^{-2}$ ), <sup>19</sup> ultrathin  $\text{Co}_3\text{S}_4$  NS ( $\eta = 650$  mV for 3.27 mA cm<sup>-2</sup>),<sup>41</sup>  $\text{Co}_3\text{O}_4$  nanorod ( $\eta = 385$  mV for 1 mA cm<sup>-2</sup>),<sup>45</sup> Co-Pi NA/Ti foam  $(\eta = 450 \text{ mV for } 10 \text{ mA cm}^{-2})$ , <sup>15</sup> Co–Bi NSs/graphene ( $\eta = 570 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ , <sup>18</sup> and Fe–Ni–P ( $\eta = 429 \text{ mV for } 14.4 \text{ mA cm}^{-2})$ 10 mA cm<sup>-2</sup>).<sup>14</sup> The detailed comparison is summarized in Table S2. Figure 4b shows the Tafel plots of all electrodes for a better understanding of the obtained catalytic behavior. The Tafel slope of 123 mV dec-1 in a neutral electrolyte for NiCo<sub>2</sub>S<sub>4</sub> NS/CC is achieved. It is the smallest value among NiCo<sub>2</sub>S<sub>4</sub> NN/CC (125 mV dec<sup>-1</sup>) and NiCo<sub>2</sub>O<sub>4</sub> NS/CC (203 mV dec<sup>-1</sup>) and comparable to that of RuO<sub>2</sub>/CC (115 mV dec<sup>-1</sup>), which in turn favors the kinetics of OER. Notably, in comparison with NiCo<sub>2</sub>O<sub>4</sub> NS/CC, the NiCo<sub>2</sub>S<sub>4</sub> NS/CC catalyst presents a lower Tafel slope value, originating from the increase in electrical conductivity as well as more plentiful electrocatalytic active sites correlated with its intrinsic activities.

We further measure the double-layer charging of electrodes via scan-rate-dependent CVs to estimate the effective surface areas for catalytic activity. The potential range in which the non-faradaic region was chosen with the potential window of 0.04 V centered at an open-circuit voltage (OCV) of each system.<sup>35</sup> The electrochemical double-layer capacitance for NiCo<sub>2</sub>S<sub>4</sub> NS/CC is 0.044 mF cm<sup>-2</sup>, whereas those for NiCo<sub>2</sub>S<sub>4</sub> NN/CC and NiCo<sub>2</sub>O<sub>4</sub> NS/CC are 0.025 and 0.023 mF cm<sup>-2</sup> respectively, indicating the rougher surface of the NiCo<sub>2</sub>S<sub>4</sub> NS/ CC electrode. It is noticeable that NiCo<sub>2</sub>S<sub>4</sub> NS/CC still possesses almost twofold higher electrochemical double-layer capacitance than NiCo2O4 NS/CC in neutral media, which is probably because of the well-aligned hierarchical NSs architecture and the formation of numerous electrochemically active sites. The TOF at the overpotential of 400 mV in neutral media is evaluated to compare the intrinsic activities of NiCo<sub>2</sub>S<sub>4</sub> NS/CC with those of other comparison electrodes. The calculated TOF for NiCo<sub>2</sub>S<sub>4</sub> NS/CC is  $9.89 \times 10^{-3}$  s<sup>-1</sup>, which is much larger than those for previously reported cobaltbased catalysts, including  $\text{Co}_3\text{S}_4$  (1.32 × 10<sup>-3</sup> s<sup>-1</sup> at  $\eta$  = 500 mV), <sup>41</sup> Co–Pi (~2 × 10<sup>-3</sup> s<sup>-1</sup> at  $\eta$  = 410 mV), <sup>46</sup> Co–Bi (1.5  $\times 10^{-3} \text{ s}^{-1} \text{ at } \eta = 400 \text{ mV}$ , <sup>47</sup> and  $\text{Co}_3\text{O}_4$  ( $\ge 0.8 \times 10^{-3} \text{ s}^{-1} \text{ at } \eta$ = 414 mV),<sup>41</sup> further suggesting the remarkable OER catalytic activity of NiCo<sub>2</sub>S<sub>4</sub> NS/CC under neutral conditions. The NiCo<sub>2</sub>S<sub>4</sub> NN/CC for which NiCo<sub>2</sub>S<sub>4</sub> NN arrays are grown on CC indicates a TOF of  $4.83 \times 10^{-3}$  s<sup>-1</sup>, implying its lower intrinsic activities compared with that of the NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrode, whereas the TOF of NiCo<sub>2</sub>O<sub>4</sub> NN/CC is calculated as  $2.47 \times 10^{-3}$  s<sup>-1</sup>, which shows the lowest value among the three electrodes. Therefore, the remarkable electrocatalytic activity of NiCo<sub>2</sub>S<sub>4</sub> NS/CC can partially originate from the higher electrochemical surface area and the direct contact between NiCo<sub>2</sub>S<sub>4</sub> NS arrays and the CC, which facilitate fast electron transfer as well as enhanced mass transportation. In addition, (i) the intrinsic electrocatalytic activity of the NiCo<sub>2</sub>S<sub>4</sub> with larger anions compared with NiCo<sub>2</sub>O<sub>4</sub> so as to expose more cation active sites; (ii) the enough void space among interconnected NiCo<sub>2</sub>S<sub>4</sub> NSs, which allows facile redox ion diffusion; (iii) the 2D morphology of NiCo<sub>2</sub>S<sub>4</sub> NSs that yields a large contact area between the catalyst and the electrolyte; and (iv) the formation of the nickel-cobalt (oxy)hydroxide active layer on its surface, which will be discussed later, all contributed to the superb performance of NiCo<sub>2</sub>S<sub>4</sub> NS/CC in the OER.

The long-term operation of the OER catalyst is a critical issue for practical application. Figure 3f exhibits the chronoamperometric (CA) curve of the NiCo<sub>2</sub>S<sub>4</sub> NS/CC measured at 1.55 V potential in alkaline media (1 M KOH). After the 29 h CA test, the electrode entirely stabilizes and retains a current density of 10 mA cm<sup>-2</sup> (without *iR* corrected) and then is reduced to 85% of its original activity over 160 h long-term operation. The inset of Figure 3f shows the linear polarization curves of NiCo<sub>2</sub>S<sub>4</sub> NS/CC before and after 1000 CV cycles with 50 mV s<sup>-1</sup> scan rate and a potential range from 1.2 to 1.7 V. Accordingly, there is no difference in the LSV curve recorded after 1000 CV cycles, indicating its high stability. Meanwhile, we detected losses of larger than 24, 29, and 47% in their current densities within 50 h for NiCo<sub>2</sub>S<sub>4</sub> NN/CC, NiCo<sub>2</sub>O<sub>4</sub> NS/CC, and RuO<sub>2</sub>/CC, respectively, at the same potential of 1.55 V in 1 M KOH solution. We also performed the durability test for same electrodes under the neutral condition. NiCo<sub>2</sub>S<sub>4</sub> NS/CC presents excellent durability for 11 h, achieving 6 mA cm<sup>-2</sup> at 1.6 V versus RHE with only 10% loss. During CA, O2 gas bubbles were visibly observed from the NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrode and dissipated quickly into the electrolyte. The NiCo<sub>2</sub>S<sub>4</sub> NN/CC and NiCo<sub>2</sub>O<sub>4</sub> NS/CC electrodes show a dramatic catalytic activity loss of 45 and 54%, respectively. Moreover, RuO<sub>2</sub>/CC almost lost its catalytic activity after 4 h durability test. The morphological robustness of NiCo<sub>2</sub>S<sub>4</sub> NS/CC was examined by post-OER FE-SEM analysis under alkaline and neutral conditions (Figure S12). Maintaining morphology with negligible damage is another convincing evidence of the structural robustness of NiCo<sub>2</sub>S<sub>4</sub> NS/CC observed in the FE-SEM micrographs.

Recently, Li et al. <sup>47</sup> reported that the nonoxide transition metal-based chalcogenides, especially cobalt selenide catalysts, usually oxidize during the OER under the basic condition and progressively transform to the corresponding TM (oxy)-hydroxides, which is proposed to be the true active species to catalyze the OER. <sup>48</sup> In the case of the  $\rm Co_3Se_4/CF$  electrode, the XPS peak intensity of Se virtually disappears after a 3 h chronopotentiometric electrolysis duration, and after 12 h,  $\rm Co_3Se_4$  is converted to CoOOH. Similarly, in our study, we

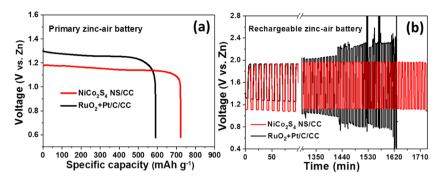


Figure 5. Zn—air battery performance: (a) specific discharge capacities of primary zinc—air batteries with  $NiCo_2S_4$  NS/CC and commercial  $RuO_2$  + Pt/C/CC air cathodes. (b) Comparative galvanostatic charge—discharge profiles of rechargeable zinc—air batteries based on  $NiCo_2S_4$  NS/CC and  $RuO_2$  + Pt/C/CC air cathodes at 5 mA cm<sup>-2</sup> in 10 min interval cycles.

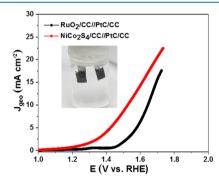
investigated the composition of the electrode after 160 h CA operation by XPS (Figure S11) to confirm the real surface species of NiCo<sub>2</sub>S<sub>4</sub> NS/CC. The XPS Ni 2p shows that Ni<sup>2+</sup> at 853.8 eV for Ni  $2p_{3/2}$  and 873.8 eV for Ni  $2p_{1/2}$  visibly disappears and the peaks located at 855.7 and 873.2 eV are assigned to Ni<sup>3+</sup> species of the nickel (oxy)hydroxide.<sup>32,49</sup> Moreover, the binding energy shift of Ni 2p for 1.4 eV reveals the occurrence of electron transfer during extended CA electrolysis. Similarly, the XPS Co  $2p_{3/2}$  peak is deconvoluted into two peaks of 780.7 and 782.3 eV, which represent the formation of Co(OH)<sub>2</sub> and CoOOH, implying the formation of a higher valence state of cobalt (Co<sup>3+</sup>). Meanwhile, the peak intensity of S 2p was weakened, whereas the two strong peaks of O 1s spectra were observed at 531.3 and 532.7 eV, indicating the O-H bond in NiCoOOH and the adsorption of  $H_2O$  on the surface of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, respectively. <sup>49-51</sup> The XPS results demonstrate that in situ electrochemical tuning of nickel-cobalt sulfide to nickel-cobalt mixed (oxy)hydroxide phase occurred, which is highly active for the OER catalytic activity attributed to the enhanced surface area and electrochemically active sites. This transformation might change the electronic states and the interactions with intermediate products during OER. Hence, it leads to the catalyst becoming more catalytic active for OER, which is also shown in other chalcogenide materials. 47,51

The post-OER durability measurement for over 11 h in the neutral medium was also carried out using XPS analysis to confirm the chemical composition (Figure S12). Similarly, in an alkaline environment, the  $\mathrm{Ni^{2+}}$  peak disappeared from the surface of catalyzed  $\mathrm{NiCo_2S_4}$  NS/CC and transformed to  $\mathrm{Ni^{3+}}$  of TM (oxy)hydroxides with binding energy values at 855.7 and 873.2 eV as well as satellite peaks at 865.1 and 879.9 eV. Also, the new peaks formed at 779.9 and 781.1 eV are also assigned to the (oxy)hydroxides phase. He is noticeable that similar phenomena of in situ electrochemical tuning for  $\mathrm{NiCo_2S_4}$  NS/CC have occurred under a neutral condition, achieving an increase in the surface area as well as electrochemically active sites for primarily improved catalytic activity for OER.

To validate the practical application of the NiCo<sub>2</sub>S<sub>4</sub> NS/CC catalyst, a primary zinc—air battery was demonstrated and fully discharged to 0.6 V at a current density of 5 mA cm<sup>-2</sup> (Figure 5a). The NiCo<sub>2</sub>S<sub>4</sub> NS/CC cathode shows an OCV of 1.18 V with a specific capacity of 722 mA h g<sup>-1</sup>, which is almost 88.1% utilization of theoretical capacity (~820 mA h g<sup>-1</sup>), whereas the commercial catalyst-based cathode shows an OCV of 1.31 V with a discharge capacity of 590 mA h g<sup>-1</sup>. Moreover, the

galvanostatic discharge—charge cycling performance was evaluated at a current density of 5 mA cm<sup>-2</sup> with a 5 min discharge followed by 5 min charge for each cycle (Figure 5b). For the initial cycle of NiCo<sub>2</sub>S<sub>4</sub> NS/CC, the rechargeable battery discharged at 1.11 V versus Zn, with the corresponding charging potential of 1.90 V giving an overall overpotential of 0.79 V, which increased only 0.04 V (1.95 V for charge and 1.12 V for discharge potential) after 30 h battery operation (173 cycles). However, in the case of RuO<sub>2</sub> + Pt/C/CC, the potential gap between charge and discharge increased continuously from 0.61 to 1.00 V even after 1350 min (135 cycles) cycling. The superior cycling durability over 173 cycles with a high discharge capacity of 722 mA h g<sup>-1</sup> indicates the excellent electrocatalytic activity and stability of NiCo<sub>2</sub>S<sub>4</sub> NS/CC for zinc—air batteries.

A two-electrode alkaline water electrolyzer was developed for full water splitting with  $NiCo_2S_4$  NS/CC and Pt/C/CC as the anode and cathode, respectively, in 1 M KOH solution (Figure 6). To achieve a current density of 10 mA cm<sup>-2</sup>, a



**Figure 6.** Overall water electrolysis: the polarization curves based on  $NiCo_2S_4/CC//PtC/CC$  and commercial  $RuO_2/CC//Pt/C/CC$  electrodes with a scan rate of 5 mV s<sup>-1</sup> in 1 M KOH solution. The inset is the photograph of the two-electrode configuration.

voltage of 1.53 V was needed for water splitting with gas evolution on both electrode surfaces, showing more advantages to split water than precious metal-based electrodes, which requires higher cell voltages of 1.64 V.

#### CONCLUSIONS

In summary, the hierarchical spinel bimetallic sulfide nanostructures in situ grown on the CC were investigated for their electrochemical properties in different pH media and evaluated for their capability in practical primary and

rechargeable zinc-air batteries. The most active NiCo<sub>2</sub>S<sub>4</sub> NS/ CC electrode can catalyze the OER at an overpotential of 260 mV at 10 mA cm<sup>-2</sup> with good durability of over 160 h operations under an alkaline condition. Moreover, the NiCo<sub>2</sub>S<sub>4</sub> NS/CC electrode still maintained its superior OER catalytic activity under the neutral condition. The enhanced intrinsic catalytic properties, morphology-based advantages of nanostructures, and the generation of the Ni-Co oxyhydroxide active layer were considered responsible for the excellent OER performance in water splitting. Especially, the in situ fabricated NiCo<sub>2</sub>S<sub>4</sub> NS/CC-integrated air cathode exhibits excellent durability and electrocatalytic activity in zinc-air batteries compared with the precious metal-based catalyst. This work supports a snapshot of the rational design and construction of nonprecious electrode materials with excellent catalytic activity and durability for the future practical system toward OER in water electrolyzers and zinc-air batteries.

### ASSOCIATED CONTENT

## S Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsomega.8b01375.

Experimental section; XRD patterns; FE-SEM images; TEM EDX pattern and mapping images; LSV curves; EIS spectra; CV curves measured in the OER region; CV curves and their corresponding charging current density versus different scan rates; postmortem FE-SEM and XPS analysis; and comparison of OER performance with the recently reported literature (PDF)

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#### Notes

The authors declare no competing financial interest.

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# REFERENCES

- (1) Mallouk, T. E. Divide and conquer. Nat. Chem. 2013, 5, 362-363.
- (2) Zou, X.; Zhang, Y. Noble metal-free hydrogen evolution catalysts for water splitting. *Chem. Soc. Rev.* **2015**, *44*, 5148–5180.
- (3) Suntivich, J.; May, K. J.; Gasteiger, H. A.; Goodenough, J. B.; Shao-Horn, Y. A perovskite oxide optimized for oxygen evolution catalysis from molecular orbital principles. *Science* **2011**, *334*, 1383–1385.
- (4) Man, I. C.; Su, H.-Y.; Calle-Vallejo, F.; Hansen, H. A.; Martínez, J. I.; Inoglu, N. G.; Kitchin, J.; Jaramillo, T. F.; Nørskov, J. K.;

Rossmeisl, J. Universality in oxygen evolution electrocatalysis on oxide surfaces. *ChemCatChem* **2011**, *3*, 1159–1165.

- (5) Vignesh, A.; Prabu, M.; Shanmugam, S. Porous LaCo1-xNixO3- $\delta$  Nanostructures as an Efficient Electrocatalyst for Water Oxidation and for a Zinc-Air Battery. ACS Appl. Mater. Interfaces **2016**, 8, 6019–6031.
- (6) Symes, M. D.; Cronin, L. Materials for Water Splitting. *Materials for a Sustainable Future*; RSC Publishing: Cambridge, UK, 2012; pp 592–614.
- (7) Hao, S.; Yang, Y. Water splitting in near-neutral media: using an Mn-Co-based nanowire array as a complementary electrocatalyst. *J. Mater. Chem. A* **2017**, *5*, 12091–12095.
- (8) Ai, L.; Tian, T.; Jiang, J. Ultrathin Graphene Layers Encapsulating Nickel Nanoparticles Derived Metal-Organic Frameworks for Highly Efficient Electrocatalytic Hydrogen and Oxygen Evolution Reactions. ACS Sustainable Chem. Eng. 2017, 5, 4771–4777.
- (9) Xu, W.; Lyu, F.; Bai, Y.; Gao, A.; Feng, J.; Cai, Z.; Yin, Y. Porous cobalt oxide nanoplates enriched with oxygen vacancies for oxygen evolution reaction. *Nano Energy* **2018**, 43, 110–116.
- (10) Zhu, G.; Ge, R.; Qu, F.; Du, G.; Asiri, A. M.; Yao, Y.; Sun, X. In situ surface derivation of an Fe-Co-Bi layer on an Fe-doped Co3O4 nanoarray for efficient water oxidation electrocatalysis under nearneutral conditions. *J. Mater. Chem. A* **2017**, *5*, 6388–6392.
- (11) Chen, W.; Liu, Y.; Li, Y.; Sun, J.; Qiu, Y.; Liu, C.; Zhou, G.; Cui, Y. *In situ* electrochemically derived nanoporous oxides from transition metal dichalcogenides for active oxygen evolution catalysts. *Nano Lett.* **2016**, *16*, 7588–7596.
- (12) Zhou, M.; Weng, Q.; Zhang, X.; Wang, X.; Xue, Y.; Zeng, X.; Bando, Y.; Golberg, D. In situ electrochemical formation of core-shell nickel-iron disulfide and oxyhydroxide heterostructured catalysts for a stable oxygen evolution reaction and the associated mechanisms. *J. Mater. Chem. A* **2017**, *5*, 4335–4342.
- (13) Sivanantham, A.; Shanmugam, S. Nickel selenide supported on nickel foam as an efficient and durable non-precious electrocatalyst for the alkaline water electrolysis. *Appl. Catal., B* **2017**, *203*, 485–493.
- (14) Ganesan, P.; Sivanantham, A.; Shanmugam, S. Inexpensive electrochemical synthesis of nickel iron sulphides on nickel foam: super active and ultra-durable electrocatalysts for alkaline electrolyte membrane water electrolysis. *J. Mater. Chem. A* **2016**, *4*, 16394–16402.
- (15) Zhang, B.; Lui, Y. H.; Ni, H.; Hu, S. Bimetallic (Fe x Ni 1-x) 2 P nanoarrays as exceptionally efficient electrocatalysts for oxygen evolution in alkaline and neutral media. *Nano Energy* **2017**, *38*, 553–560
- (16) Du, C.; Yang, L.; Yang, F.; Cheng, G.; Luo, W. Nest-like NiCoP for highly efficient overall water splitting. ACS Catal. 2017, 7, 4131–4137
- (17) Xie, L.; Zhang, R.; Cui, L.; Liu, D.; Hao, S.; Ma, Y.; Du, G.; Asiri, A. M.; Sun, X. High-performance electrolytic oxygen evolution in neutral media catalyzed by a cobalt phosphate nanoarray. *Angew. Chem.* **2017**, *129*, 1084–1088.
- (18) Lu, Z.; Xu, W.; Zhu, W.; Yang, Q.; Lei, X.; Liu, J.; Li, Y.; Sun, X.; Duan, X. Three-dimensional NiFe layered double hydroxide film for high-efficiency oxygen evolution reaction. *Chem. Commun.* **2014**, 50, 6479–6482.
- (19) Jiang, J.; Zhang, A.; Li, L.; Ai, L. Nickel-cobalt layered double hydroxide nanosheets as high-performance electrocatalyst for oxygen evolution reaction. *J. Power Sources* **2015**, 278, 445–451.
- (20) Jiang, J.; Liu, Q.; Zeng, C.; Ai, L. Cobalt/molybdenum carbide@N-doped carbon as a bifunctional electrocatalyst for hydrogen and oxygen evolution reactions. *J. Mater. Chem. A* **2017**, 5, 16929–16935.
- (21) Chen, P.; Xu, K.; Zhou, T.; Tong, Y.; Wu, J.; Cheng, H.; Lu, X.; Ding, H.; Wu, C.; Xie, Y. Strong-coupled cobalt borate nanosheets/graphene hybrid as electrocatalyst for water oxidation under both alkaline and neutral conditions. *Angew. Chem., Int. Ed.* **2016**, 55, 2488–2492.

(22) Cai, P.; Huang, J.; Chen, J.; Wen, Z. Oxygen-Containing Amorphous Cobalt Sulfide Porous Nanocubes as High-Activity Electrocatalysts for the Oxygen Evolution Reaction in an Alkaline/Neutral Medium. *Angew. Chem., Int. Ed.* **2017**, *56*, 4858–4861.

- (23) Liu, J.; Zhu, D.; Ling, T.; Vasileff, A.; Qiao, S.-Z. S-NiFe 2 O 4 ultra-small nanoparticle built nanosheets for efficient water splitting in alkaline and neutral pH. *Nano Energy* **2017**, *40*, 264–273.
- (24) Prabu, M.; Ketpang, K.; Shanmugam, S. Hierarchical nanostructured NiCo2O4 as an efficient bifunctional non-precious metal catalyst for rechargeable zinc-air batteries. *Nanoscale* **2014**, *6*, 3173–3181.
- (25) Li, Y.; Zhou, W.; Dong, J.; Luo, Y.; An, P.; Liu, J.; Wu, X.; Xu, G.; Zhang, H.; Zhang, J. Interface engineered in situ anchoring of Co9S8 nanoparticles into a multiple doped carbon matrix: highly efficient zinc-air batteries. *Nanoscale* **2018**, *10*, 2649–2657.
- (26) Meng, T.; Qin, J.; Wang, S.; Zhao, D.; Mao, B.; Cao, M. In situ coupling of Co0.85Se and N-doped carbon via one-step selenization of metal—organic frameworks as a trifunctional catalyst for overall water splitting and Zn—air batteries. *J. Mater. Chem. A* **2017**, *5*, 7001—7014.
- (27) Wu, X.; Han, X.; Ma, X.; Zhang, W.; Deng, Y.; Zhong, C.; Hu, W. Morphology-Controllable Synthesis of Zn-Co-Mixed Sulfide Nanostructures on Carbon Fiber Paper Toward Efficient Rechargeable Zinc-Air Batteries and Water Electrolysis. ACS Appl. Mater. Interfaces 2017, 9, 12574—12583.
- (28) Wang, H.-F.; Tang, C.; Wang, B.; Li, B.-Q.; Zhang, Q. Bifunctional Transition Metal Hydroxysulfides: Room-Temperature Sulfurization and Their Applications in Zn-Air Batteries. *Adv. Mater.* **2017**, *29*, 1702327.
- (29) Xue, Y.; Zuo, Z.; Li, Y.; Liu, H.; Li, Y. Graphdiyne-Supported NiCo2 S4 Nanowires: A Highly Active and Stable 3D Bifunctional Electrode Material. *Small* **2017**, *13*, 1700936.
- (30) Miao, J.; Xiao, F.-X.; Yang, H. B.; Khoo, S. Y.; Chen, J.; Fan, Z.; Hsu, Y.-Y.; Chen, H. M.; Zhang, H.; Liu, B. Hierarchical Ni-Mo-S nanosheets on carbon fiber cloth: a flexible electrode for efficient hydrogen generation in neutral electrolyte. *Sci. Adv.* **2015**, *1*, 1500259.
- (31) Sivanantham, A.; Ganesan, P.; Shanmugam, S. Hierarchical NiCo2S4Nanowire Arrays Supported on Ni Foam: An Efficient and Durable Bifunctional Electrocatalyst for Oxygen and Hydrogen Evolution Reactions. *Adv. Funct. Mater.* **2016**, *26*, 4661–4672.
- (32) Ma, L.; Hu, Y.; Chen, R.; Zhu, G.; Chen, T.; Lv, H.; Wang, Y.; Liang, J.; Liu, H.; Yan, C.; Zhu, H.; Tie, Z.; Jin, Z.; Liu, J. Self-assembled ultrathin NiCo 2 S 4 nanoflakes grown on Ni foam as high-performance flexible electrodes for hydrogen evolution reaction in alkaline solution. *Nano Energy* **2016**, *24*, 139–147.
- (33) Tian, J.; Liu, Q.; Liang, Y.; Xing, Z.; Asiri, A. M.; Sun, X. FeP Nanoparticles Film Grown on Carbon Cloth: An Ultrahighly Active 3D Hydrogen Evolution Cathode in Both Acidic and Neutral Solutions. ACS Appl. Mater. Interfaces 2014, 6, 20579–20584.
- (34) McCrory, C. C. L.; Jung, S.; Peters, J. C.; Jaramillo, T. F. Benchmarking heterogeneous electrocatalysts for the oxygen evolution reaction. *J. Am. Chem. Soc.* **2013**, *135*, 16977–16987.
- (35) Sun, M.; Tie, J.; Cheng, G.; Lin, T.; Peng, S.; Deng, F.; Ye, F.; Yu, L. In situ growth of burl-like nickel cobalt sulfide on carbon fibers as high-performance supercapacitors. *J. Mater. Chem. A* **2015**, 3, 1730–1736.
- (36) Wang, J.-G.; Jin, D.; Zhou, R.; Shen, C.; Xie, K.; Wei, B. Onestep synthesis of NiCo 2 S 4 ultrathin nanosheets on conductive substrates as advanced electrodes for high-efficient energy storage. *J. Power Sources* **2016**, *306*, 100–106.
- (37) Liu, D.; Lu, Q.; Luo, Y.; Sun, X.; Asiri, A. M. NiCo2S4nanowires array as an efficient bifunctional electrocatalyst for full water splitting with superior activity. *Nanoscale* **2015**, *7*, 15122–15126.
- (38) Li, Y.; Hasin, P.; Wu, Y. NixCo3-xO4 Nanowire Arrays for Electrocatalytic Oxygen Evolution. *Adv. Mater.* **2010**, 22, 1926–1929. (39) Liu, J.; Wang, J.; Zhang, B.; Ruan, Y.; Lv, L.; Ji, X.; Xu, K.; Miao, L.; Jiang, J. Hierarchical NiCo2S4@NiFe LDH Hetero-

structures Supported on Nickel Foam for Enhanced Overall-Water-Splitting Activity. ACS Appl. Mater. Interfaces 2017, 9, 15364–15372.

- (40) Liu, Y.; Xiao, C.; Lyu, M.; Lin, Y.; Cai, W.; Huang, P.; Tong, W.; Zou, Y.; Xie, Y. Ultrathin Co3S4Nanosheets that Synergistically Engineer Spin States and Exposed Polyhedra that Promote Water Oxidation under Neutral Conditions. *Angew. Chem.* **2015**, *127*, 11383–11387.
- (41) Trotochaud, L.; Ranney, J. K.; Williams, K. N.; Boettcher, S. W. Solution-Cast Metal Oxide Thin Film Electrocatalysts for Oxygen Evolution. *J. Am. Chem. Soc.* **2012**, *134*, 17253–17261.
- (42) Zhang, Z.; Wang, X.; Cui, G.; Zhang, A.; Zhou, X.; Xu, H.; Gu, L. NiCo2S4 sub-micron spheres: an efficient non-precious metal bifunctional electrocatalyst. *Nanoscale* **2014**, *6*, 3540–3544.
- (43) Suntivich, J.; May, K. J.; Gasteiger, H. A.; Goodenough, J. B.; Shao-Horn, Y. A Perovskite Oxide Optimized for Oxygen Evolution Catalysis from Molecular Orbital Principles. *Science* **2011**, *334*, 1383–1385
- (44) Ramsundar, R. M.; Debgupta, J.; Pillai, V. K.; Joy, P. A. Co3O4 Nanorods-Efficient Non-noble Metal Electrocatalyst for Oxygen Evolution at Neutral pH. *Electrocatalysis* **2015**, *6*, 331–340.
- (45) Surendranath, Y.; Kanan, M. W.; Nocera, D. G. Mechanistic Studies of the Oxygen Evolution Reaction by a Cobalt-Phosphate Catalyst at Neutral pH. *J. Am. Chem. Soc.* **2010**, *132*, 16501–16509.
- (46) Esswein, A. J.; Surendranath, Y.; Reece, S. Y.; Nocera, D. G. Highly active cobalt phosphate and borate based oxygen evolving catalysts operating in neutral and natural waters. *Energy Environ. Sci.* **2011**, *4*, 499–504.
- (47) Li, W.; Gao, X.; Xiong, D.; Wei, F.; Song, W.-G.; Xu, J.; Liu, L. Hydrothermal Synthesis of Monolithic Co3 Se4 Nanowire Electrodes for Oxygen Evolution and Overall Water Splitting with High Efficiency and Extraordinary Catalytic Stability. *Adv. Energy Mater.* **2017**, *7*, 1602579.
- (48) Mansour, A. N.; Melendres, C. A. Characterization of Electrochemically Prepared  $\gamma$ -NiOOH by XPS. *Surf. Sci. Spectra* **1994**, 3, 271–278.
- (49) Yang, J.; Liu, H.; Martens, W. N.; Frost, R. L. Synthesis and characterization of cobalt hydroxide, cobalt oxyhydroxide, and cobalt oxide nanodiscs. *J. Phys. Chem. C* **2010**, *114*, 111–119.
- (50) McIntyre, N. S.; Cook, M. G. X-ray photoelectron studies on some oxides and hydroxides of cobalt, nickel, and copper. *Anal. Chem.* 1975, 47, 2208.
- (51) Chen, W.; Wang, H.; Li, Y.; Liu, Y.; Sun, J.; Lee, S.; Lee, J.-S.; Cui, Y. In Situ Electrochemical Oxidation Tuning of Transition Metal Disulfides to Oxides for Enhanced Water Oxidation. *ACS Cent. Sci.* **2015**, *1*, 244–251.